

Photoconductivity and surface chemical analysis of ZnO thin films deposited by solution-processing techniques for nano and microstructure fabrication

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Abstract: The fabrication of zinc oxide (ZnO) from inexpensive solution-processing techniques, namely, electrochemical deposition and electrospinning were explored on various conducting and mesoporous semiconducting surfaces. Optimised conditions were derived for template- and self-assisted nano/micro structures and composites. ZnO thin films were annealed at a fixed temperature under ambient conditions and characterised using physical and optical techniques. The photocurrent response in the UV region shows a fast rise and double decay behaviour with a fast component followed by a slow oscillatory decay. Photocurrent results were correlated with surface chemical analysis from X-ray photoelectron spectroscopy. Various characterisation details reveal the importance of fabrication parameter optimisation for useful low-cost optoelectronic applications.

Key words: zinc oxide; electrochemical deposition; surface analysis; X-ray photoelectron spectroscopy; photocurrent response; optoelectronic applications

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1. Introduction

ZnO has attracted increasing attention as a potential material for optoelectronic devices such as low threshold blue/UV lasers, solar cells, LEDs, sensors, display devices and photodetectors^[1–3]. In recent years, various attempts were made to fabricate nano/mesa-scaled ZnO for further enhancing opto/electrical performance. While many top-down fabrication approaches were followed, solution-based techniques turned out to be the most efficient and low-cost for the production of high quality nano and micro structured ZnO thin films^[4, 5]. Electrochemical deposition is a true bottom-up technique, with the convenience of completely filling the interstitial spaces of templates, such as polymer microsphere templates, liquid crystal templates and porous (such as porous alumina) templates, from metal to semiconductor varieties^[6, 7]. Similarly, electrospinning has been widely recognised as an electrohydrodynamic method to produce nano to micro sized fibers from solutions containing the desired materials^[8, 9]. However, the optoelectronic properties of ZnO are critically affected by the preparation conditions such as the fabrication method, types of substrates, thickness and annealing conditions. Therefore, it is necessary to optimise the fabrication conditions of the aforementioned techniques for desired applications.

The direct electrochemical deposition of semiconductors, such as ZnO, on a silicon substrate is difficult due to conductivity issues as well as a large mismatch in thermal expansion coefficients and the high reactivity of silicon toward oxygen^[10–14]. Porous silicon (PS) has been established as the most fascinating material for diverse optoelectronic applications^[15–17], especially for photonic composites where the nano sized pores can be filled with different materials using simple solution-processing techniques^[18, 19].

In this paper, we have explored various solution-processing techniques to produce ZnO thin films on conducting (ITO) and silicon substrates along with ZnO-porous silicon composite films. All these samples obtained from different methods and chemical recipes were annealed under fixed ambient conditions and characterised by scanning electron microscopy (SEM), X-ray diffraction (XRD), X-ray photoelectron spectroscopy (XPS), optical absorption, photoluminescence (PL) and ultraviolet photocurrent response. All films obtained from various chemical recipes were essentially in the form of Zn(OH)₂ after preparation, and to obtain ZnO, the samples were annealed at 400 °C for 12 h under ambient conditions. Various characterisations reveal that the fabrication conditions and intrinsic defects of ZnO play a vital role in optoelectronic performance.

2. Experimental details

ZnO nano and micro structures were fabricated from template-assisted electrochemical deposition, and electrospinning methods. In electrochemical deposition, two aqueous electrolyte solutions were employed, namely, (1) 0.1M Zn(NO₃)₂ and (2) a mixture of 5 mM ZnCl₂ with 0.1 M KCl on both the ITO and silicon substrates using a potentiostatic three-electrode method^[7, 20, 21].

Similarly porous silicon is used as a template to fabricate nano-scaled ZnO. PS with different porosities were made by electrochemical etching using a p(100) type silicon wafer, as detailed in Ref. [18]. ZnO-porous silicon composites were prepared by electrochemical deposition of ZnO into the pores of PS, from a ZnCl₂ electrolyte as mentioned above. In this paper, we have shown the results of the ZnO-PS nanocomposites, where ZnO is electrodeposited into the pores of the 70% porous

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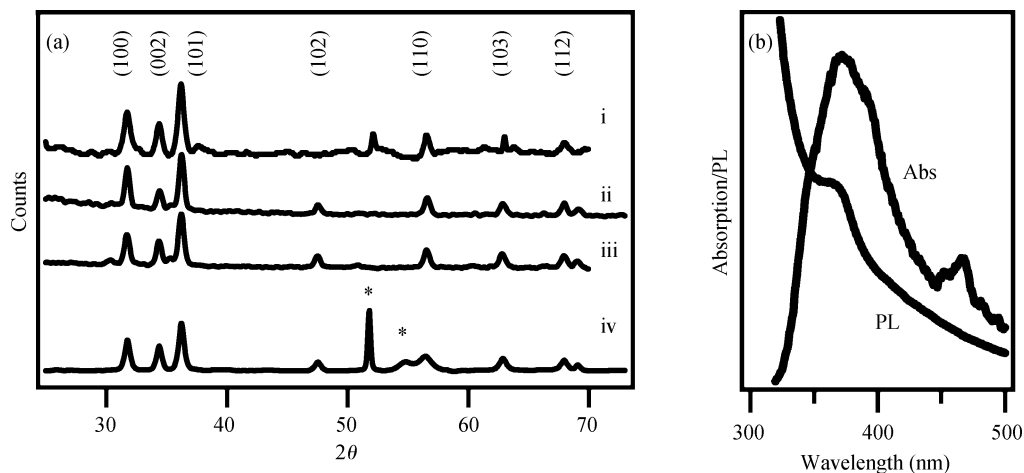


Fig. 1. (a) XRD patterns of ZnO (i) electrodeposited from ZnCl_2 and (ii) electrodeposited from $\text{Zn}(\text{NO}_3)_2$ solutions, (iii) electro-spun ZnO from $\text{Zn}(\text{Ac})_2$ solution and (iv) ZnO-PS nanocomposite. The asterisk (*) indicates XRD peaks of silicon. XRD patterns are shifted along the y-axis for clarity. (b) Absorption and photoluminescence spectra of electro-spun ZnO fibers.

ity PS of thickness ~ 150 nm on the silicon substrate. To ensure complete wetting of the pores, the PS was soaked in the electrolyte for about 1 h.

All films obtained from various chemical recipes were essentially in the form of $\text{Zn}(\text{OH})_2$ and to obtain ZnO, the samples were annealed at 400°C for 12 h under ambient conditions.

The crystalline phases of the thin films were identified by an X-ray diffraction (XRD) technique using a $\text{CuK}\alpha$ source ($\lambda = 1.54 \text{ \AA}$). The chemical state of the different elements present in the thin films was analyzed by X-ray photoelectron spectroscopy (XPS) using a $\text{Mg K}\alpha$ source ($h\nu = 1253.6 \text{ eV}$). Prior to these measurements, Argon etching for a 2 min duration was performed to remove a few atomic layers of the surface to avoid any surface contamination. The binding energy of the XPS spectra has been calibrated by taking the C1s peak (284.6 eV) as a reference. Steady-state and transient photocurrent response studies of ZnO films were investigated by the three-electrode wet-contact method^[22]. A commercial Si detector (UDT model No. 260) was used as a photocurrent response reference. For photocurrent measurements, a wet-contact three-electrode method was used wherein an aqueous solution of 0.1 M KI acts as an electrolyte. A saturated Calomel electrode, a platinum mesh and a ZnO film deposited on substrates were used as the reference, counter and working electrodes respectively. A nitrogen laser with $\sim 2 \text{ mW}$ power (337 nm wavelength, 100 Hz pulsed frequency and 5 ns pulse width) was used as a UV irradiation and photocurrent was monitored by computer controlled potentiostat and a digital oscilloscope. To measure the photocurrent in the spectral domain, Xenon (75 W) attached to a monochromator is used as a photoexcitation source. PL measurements were carried out using a monochromator and a PMT, using a nitrogen laser (337 nm) as an excitation source.

3. Results and discussion

3.1. Structural analysis

The obtained ZnO thin films from various solution processing methods show considerable variation in structural, op-

tical and optoelectronic properties on deposition parameters and conditions^[7]. Since the annealing condition for all the films was same (400°C for 12 h in air), in the present study the structural and optical properties of various films are compared and analyzed. As shown in Fig. 1, the XRD patterns were recorded for all annealed ZnO thin films obtained from (1) electrodeposition from ZnCl_2 onto a silicon substrate, (2) electrodeposition from $\text{Zn}(\text{NO}_3)_2$ solutions onto an ITO substrate, (3) electro-spun ZnO from a $\text{Zn}(\text{Ac})_2$ solution onto an ITO substrate, and (4) a ZnO-PS nanocomposite. Asterisks (*) indicate the XRD peaks of silicon. All the films were showing strong XRD peaks of wurtzite crystalline structure, with traces of substrate (Si) related diffraction peaks.

In our previous communication^[7], we have shown photoluminescence (PL) studies and reported that the ZnO fabricated from various electrochemical deposition parameters show different PL: exciton (in the UV region) related PL as well as deep level defects related PL (in blue-green region). In particular, the deep level defects were attributed to the oxygen vacancy defects and interstitial Zn ions. While ZnO obtained from electrodeposition shows both exciton as well as defect related PL, the ZnO obtained from electro-spinning shows dominant excitonic related features in both absorption as well as PL, indicating the minimum influence of defects (Fig. 1(b)).

In brief, the ZnO obtained from different solution processing methods shows widely different PL properties. In order to establish the ZnO as a multi-functional material, it is worthwhile to study and understand the photocurrent response of these samples, especially the generation and recombination of photo-carriers. Therefore, such systematic studies were performed and presented as follows.

3.2. Photocurrent studies

Steady-state and transient photocurrent measurements were performed on all ZnO films made from different solution processing techniques. Here the ZnO films were exposed to a UV laser (2 mW , 337 nm) and the photocurrent was monitored using the wet electrode method. Figure 2(a) shows the normalised photocurrent response, when the illuminated light

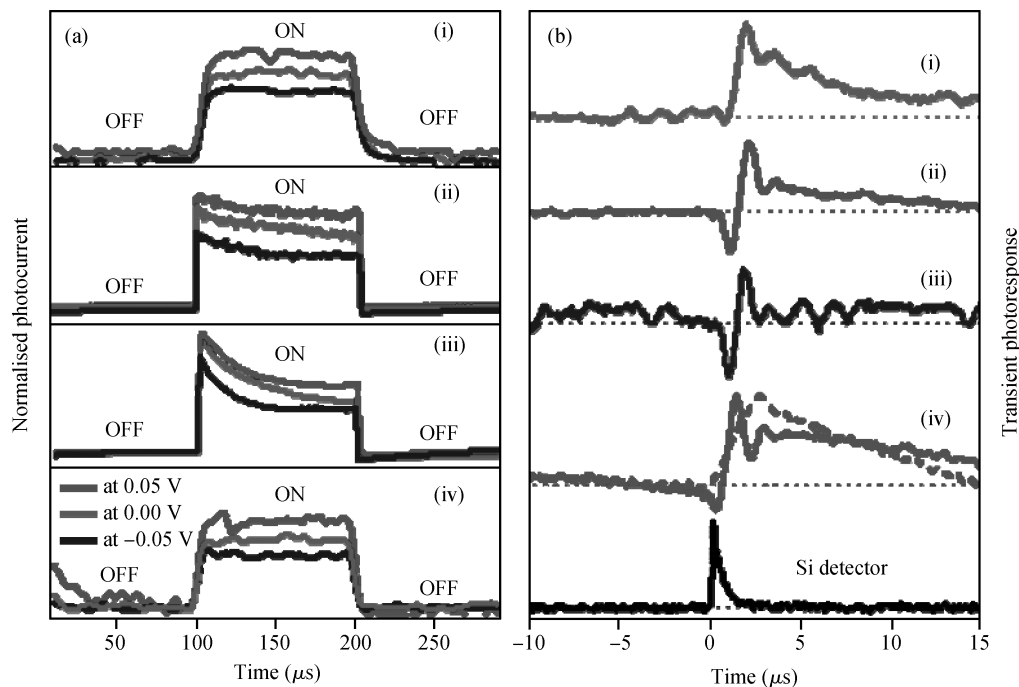


Fig. 2. (a) Normalised photocurrent ON–OFF response of ZnO films prepared from various methods. (b) Transient photocurrent response of ZnO deposited from various methods, the commercial Si detector response is shown as a reference. Labels i, ii, iii and iv represent the same as Fig. 1(a). The transient response for pure PS, overlapping ZnO-PS (dashed curve in Fig. 2(b) (iv)), is shown for comparison. All the graphs in Fig. (b) are shifted along the y-axis for clarity.

Table 1. Photocurrent response rise time, fall and slow fall times and efficiencies (QE) and estimated percentage of oxygen vacancies from the XPS analysis of ZnO fabricated from various methods. The data of ZnO prepared from various other methods reported in the literature are also given for comparison.

ZnO type	Rise time (μs)	Fast fall (μs)	Slow fall (μs)	QE	Percentage of oxygen vacancies from XPS (%)
Electrochemical deposited ZnO from ZnCl_2 sol.	1.37	1.8	27	07	26
Electrochemical deposited ZnO from ZnNO_3 sol.	1.23	1.55	20	10	29
Electrospinned ZnO from ZnAc sol.	1.1	1.3	15	41	31
ZnO-PS nanocomposite	0.9	1.2	39	05	50
Porous silicon (PS)	1.7	9.0	23	0.25	—
ZnO from RF sputtering ^[23]	0.1	0.5	1.5	—	—
$\text{Mg}_{0.34}\text{Zn}_{0.66}\text{O}$ from pulsed laser deposition ^[24]	0.008	~ 1.4	30	—	—
ZnO from MOCVD ^[25]	0.012	0.005	5×10^{-6}	—	—
ZnO from RF sputtering ^[26]	0.02	10	25	—	—

was switched ON and OFF at different bias conditions. The data represents the ZnO thin films obtained from (1) electrochemical deposition from ZnCl_2 , (2) electrochemical deposition from $\text{Zn}(\text{NO}_3)_2$ solutions, (3) electrospinned film from a $\text{Zn}(\text{Ac})_2$ solution and (4) a ZnO-PS nanocomposite. As seen, the photocurrent varies linearly with the increase in the bias voltage. While all films show more or less similar responses, the photocurrent efficiencies measured under similar conditions are different. Here, the photocurrent efficiency is defined as $\text{QE} = (I_{\text{photo}} - I_{\text{dark}}) / I_{\text{dark}}$, where the I_{dark} and I_{photo} are the photocurrents when the light is OFF and ON, respectively. The QE for ZnO obtained from electrospinned film from $\text{Zn}(\text{Ac})_2$ solution is 41 whereas for ZnO-PS the nanocomposite is only 5.

To study the transient photocurrent behaviour, the resultant photocurrent response obtained from the excitation of a

nanosecond UV laser (337 nm, 5 ns, 100 Hz) was monitored through a digital oscilloscope using a commercial silicon detector as a photocurrent response reference (Fig. 2(b)). Essentially all films show a fast photocurrent rise followed by an oscillatory decay. The photocurrent decay is composed of two components: a fast fall followed by slow fall of an oscillatory nature. The transient response is fitted using a bi-exponential function, $I = I_0 + A \exp(t/\tau_1) + B \exp(t/\tau_2)$. Both the rise and fall time values obtained from such fittings are given in Table 1. A similar fast decay followed by oscillatory decay behaviour has been reported previously for ZnO from RF sputtering^[23, 24], ZnO alloyed with MgO from pulsed laser deposition^[25], and epitaxial ZnO obtained from MOCVD^[26]. The present decay times were compared with these reported values in Table 1.

In general, the order of rise and initial fast fall times of var-

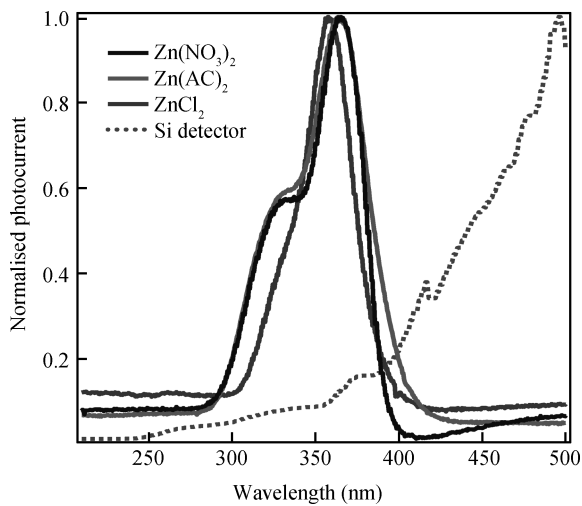


Fig. 3. Wavelength dependent photocurrent response of ZnO electro deposited thin films obtained from $\text{Zn}(\text{NO}_3)_2$, $\text{Zn}(\text{Ac})_2$ and ZnCl_2 solutions. The bias voltage is set at 0 V. The spectral photocurrent response of the commercial Si detector is also included for comparison.

ious ZnO shows the following trend: ZnO-PS nanocomposite > ZnO from $\text{Zn}(\text{Ac})_2$ > ZnO from $\text{Zn}(\text{NO}_3)_2$ > ZnO from ZnCl_2 . It is interesting to compare these results with the QE values obtained earlier, where the QE value of ZnO from $\text{Zn}(\text{Ac})_2$ is the highest and ZnO-PS is the lowest.

The photoconductivity phenomena in ZnO are generally related to either surface or to bulk related processes. The surface related process are due to adsorption and desorption of the chemisorbed oxygen on the surface of ZnO. Upon photo excitation, using light energy greater than the band gap of the material, three types of photocurrent processes are expected: (1) chemisorption of O_2 , (2) desorption of O_2 and (3) recombination of the electron-hole pair. The dominant process among these three defines the resultant nature of photocurrent. The holes eventually react with chemisorbed oxygen and release O_2 ($\text{O}_2^- + \text{h}^+ \rightarrow \text{O}_2(\text{g})$ (desorption)) leaving behind electrons, which take part in photocurrent. Further, oxygen molecules are chemisorbed on the surface of ZnO and capture the electrons as $\text{O}_2(\text{g}) + \text{e}^- \rightarrow \text{O}_2^-$ (adsorbed species)^[27, 28]. Therefore, the transient photocurrent response rise time is due to the desorption of O_2 . Whereas, the photocurrent fast fall is due to the transit time taken by the carriers, which is inversely proportional to the electron mobility and bias voltage^[22]. The oscillatory slow decay phenomenon is the result of persistent photoconductivity effects attributed to the presence of oxygen-related hole-trap states at the surface, which prevent charge-carrier recombination and prolongs the photocarrier lifetime. Similar behaviour has been observed for the ZnO fabricated from various other methods^[23–25]. While the ZnO grown from various aforementioned methods shows a good crystalline quality, the origin of oxygen defects on the surfaces could be from both deposition conditions as well as annealing in an air atmosphere. However, in the present case, though solution processing recipes are different, the obtained ZnO may be compared since the annealing conditions for all the ZnO films are the same. Hence, these results broadly suggest that the photocurrent response of ZnO could be related to its structural and

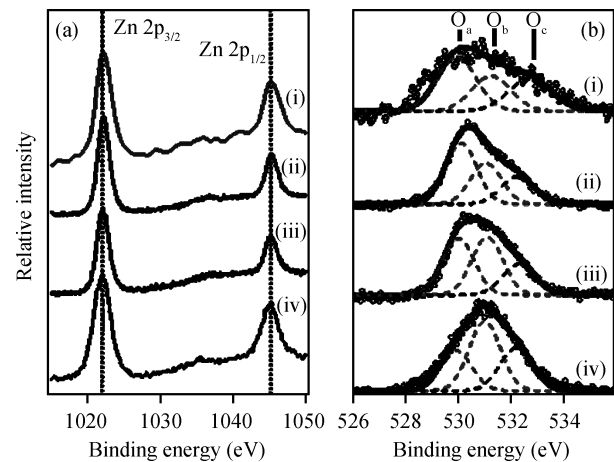


Fig. 4. (Colour online) XPS spectra of (a) Zn2p core levels and (b) O1s core level with Gaussian fits (dotted lines) for O_a , O_b and O_c . Labels i, ii, iii and iv represent the same as Fig. 1(a). Spectra are shifted along the Y-axis for clarity.

surface properties, essentially to the process induced intrinsic defects of ZnO^[29, 30]. Figure 3 shows the wavelength (200–500 nm) dependent photocurrent characteristics of ZnO thin films fabricated from various precursors. The spectral responsivity shows a distinct broad peak centred at about 360 nm, which correspond to the band gap of ZnO^[7, 27, 32].

3.3. Surface analysis

To get more information about the surface composition and the intrinsic defects (Zn interstitials and oxygen vacancies), X-ray photoelectron spectroscopy (XPS) measurements were carried out on all these samples. Figure 4(a) shows the XPS spectra of Zn2p core levels, where the peaks at 1021.8 and at 1044.9 eV correspond to $\text{Zn}2\text{p}_{3/2}$ and $\text{Zn}2\text{p}_{1/2}$ core levels respectively which confirm the presence of Zn in a +2 state in the matrix. Figure 4(b) shows that the O1s peak is asymmetric and can be fitted by three nearly Gaussian curves centred at 529.9, 531 and 532.2 eV. The peak at 532.2 eV (O_c) is usually attributed to the presence of loosely bound oxygen on the surface of the ZnO, belonging to a specific species, e.g., adsorbed H_2O or adsorbed O_2 . The medium binding energy component 531(O_b) is associated with O^{2-} ions in oxygen deficient regions (oxygen vacancies) within the matrix of ZnO. The component on the low binding energy side of the O1s spectrum at 529.9 eV (O_a) is due to the oxygen in the ZnO matrix^[31, 32]. The oxygen vacancies estimated from the XPS results follow the trend: ZnO-PS nanocomposite > ZnO from $\text{Zn}(\text{Ac})_2$ > ZnO from $\text{Zn}(\text{NO}_3)_2$ > ZnO from ZnCl_2 . These results follow the same trend as the photocurrent response results. Therefore, the photocurrent response of ZnO could be broadly related to the oxygen vacancies, especially as the rise time and fast decay of photocurrent appears to be inversely dependent on the oxygen vacancies. Such fast photocurrent response times are much more favourable for the Schottky type of UV detectors^[26]. In future, it would be worthwhile extending the study to further investigate the effect of annealing conditions and surface morphology of various nanostructured films.

4. Conclusion

Several solution-processing inexpensive deposition techniques, such as template-assisted electrochemical deposition and electrospinning have been explored and optimised conditions were derived to obtain high-quality wurtzite crystalline ZnO. Furthermore, the optical and physical properties of thin films obtained from these deposition techniques were investigated. While all the samples show a wurtzite crystalline structure, their optical properties are significantly different. These films show a typical fast photocurrent response component of rise time about 1 μ s and fall time of 2 μ s with persistent photoconductive behaviour. XPS surface analysis observations suggest a possible correlation between oxygen related defects to the photocurrent measurements. The low-cost methodology and multi-functional optical properties (fast photocurrent response and exciton PL) are useful for low-cost UV optoelectronic devices.

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